

*Full Length Research Paper*

# Effect of aging time and calcination temperature on the cerium oxide nanoparticles synthesis via reverse co-precipitation method

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**In the present work, the cerium nitrate hexa-hydrate and ammonium hydroxide, used as the starting materials and the weakly agglomerated cerium oxide nanoparticles was synthesized by simple and cost effective reverse co-precipitation method. The effect of aging time and calcination temperature on structural properties of synthesized nanopowder was investigated by X-ray diffractometry (XRD), simultaneous thermal analysis (STA), Fourier transform infrared (FT-IR), scanning electron microscopy (SEM) and transmission electron spectroscopy (TEM) analytical methods. The investigation showed that aging of the precipitated precursors has pronounced effect on the decreasing of the agglomeration, increase in the crystallinity and the crystallite size of the final product obtained after calcination. Also, increase in the calcination temperature led to the appreciable increase in the crystallite size as well as the crystallinity.**

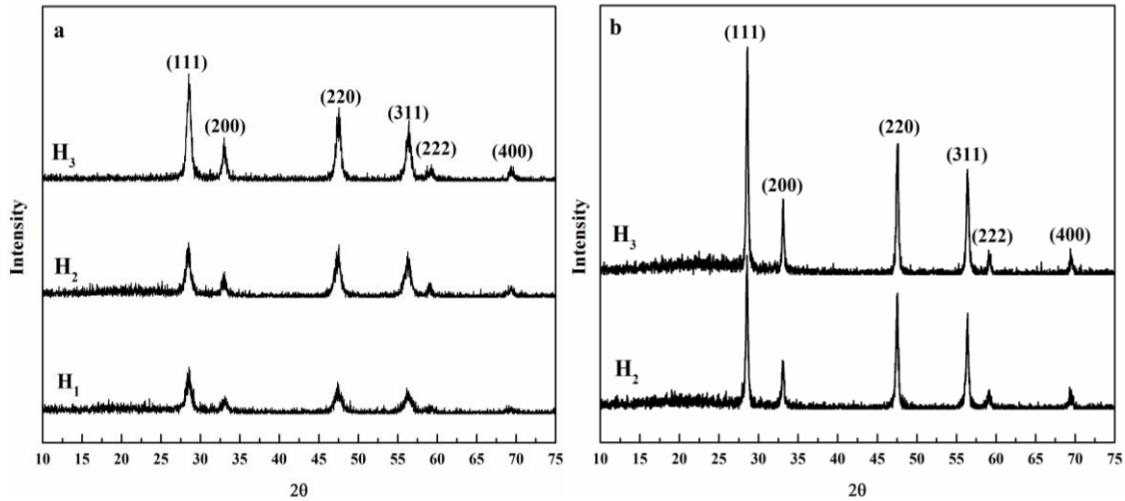
**Key words:** Aging time, ammonium hydroxide, calcination temperature, ceria nanoparticles, reverse co-precipitation.

## INTRODUCTION

Ceria ( $\text{CeO}_2$ ) is a cubic fluorite-type structured ceramic material that does not show any known crystallographic change from room temperature up to its melting point ( $2700^\circ\text{C}$ ) (Corradi et al., 2006; Li et al., 2002; Hassanzadeh-tabrizi et al., 2010). In recent years, nanocrystalline cerium oxide ( $\text{CeO}_2$ ) particles have been extensively studied owing to their potential uses in many applications, such as UV absorbents and filters (Cheviré et al., 2006; Souza et al., 2010), gas sensors (Izu et al., 2003; Jasinski et al., 2003), electrolytes in the fuel cell technology (Lapa et al., 2010; Matovic et al., 2009), water-gas shift catalysts (Gorte and Zhao, 2005), polishers for chemical mechanical planarization (CMP) (Xin et al., 2010; Lee et al., 2009), ceramic pigments (Llusar et al., 2010; Matovic et al., 2010), etc. Most of the applications require the use of non-agglomerated nanoparticles, as aggregated nanoparticles lead to inhomogeneous mixing and poor sinterability. However,

nanocrystallites with a primary particle size  $<5$  nm have a stronger tendency to agglomerate, making processing very difficult. Also, the benefits expected from highly crystalline nanoparticles are easily lost during the manufacture of components unless weakly agglomerated nanoparticles can be produced (Tok et al., 2007). Accordingly, preparation of ultrafine  $\text{CeO}_2$  powders without agglomeration has been intensively investigated.  $\text{CeO}_2$  nanopowders can be synthesized by different methods, such as, hydrothermal (Dos Santos et al., 2008), mechanochemical (Zec et al., 2009), combustion synthesis (Mukherjee et al., 2008), sol-gel (Tao et al., 2009), microemulsion (Chandradass et al., 2009), co-precipitation (Godinho et al., 2007) and spray-pyrolysis (Yao and Xie, 2007). Compared with these methods, co-precipitation method is one of the most promising techniques because of the inexpensive starting materials, a simple synthesis process with reproducible results and commonly available apparatus (Truffault et al., 2010; Hassanzadeh-tabrizi et al., 2010). Although,  $\text{CeO}_2$  nanoparticles have been extensively studied by precipitation method but most of the previous reports were focused on

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**Figure 1.** The X-ray diffraction pattern of ceria nanoparticles calcined at (a) 600°C and (b) 800°C.

on direct precipitation of ceria (Yamashita et al., 2002; Zhou et al., 2002; Uekawa et al., 2002).

In this work, the ceria nanoparticles were synthesized via reverse co-precipitation method, using cerium nitrate hexa-hydrate and ammonium hydroxide as precipitation agent. The purpose of this work was to investigate the aging time and calcination temperature effect on the structural characteristics of the final product.

## MATERIALS AND METHODS

In this study, cerium nitrate hexa-hydrate ( $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ; extra pure, Merck Chemical Company, Inc.) and ammonium hydroxide ( $(\text{NH}_4)\text{OH}$ ; extra pure, Merck Chemical Company, Inc.) were used as the starting materials. The cerium nitrate solution with 0.1 M concentration was prepared and added dropwise at the speed of 3 to 4 ml/min into the 1 M ammonium hydroxide stirring solution until the precipitation process was completed. The suspensions were aged at room temperature for 1, 6 and 12 h, respectively and were then centrifuged to collect precursor. The precursors were then thoroughly washed 3 times with distilled water and were rinsed with ethanol, respectively. They were then dried at 80°C for 12 h and calcined at 600 and 800°C for 2 h to yield the cerium oxide nanoparticles. The synthesized samples obtained at various aging times (1, 6 and 12 h) were named  $\text{H}_{1-600}$ ,  $\text{H}_{6-600}$  and  $\text{H}_{12-600}$ , after calcination at 600°C and  $\text{H}_{1-800}$ ,  $\text{H}_{6-800}$  and  $\text{H}_{12-800}$  after calcination at 800°C, respectively.

Phase identification was done via X-ray diffractometry (XRD) on a Philips X-ray diffractometer (PW3710) operating at 20 kV/10 mA using nickel-filtered  $\text{CuK}\alpha$  radiation in the range of  $2\theta = 10$  to  $80^\circ$ . Crystallite sizes ( $D_{\text{XRD}}$ ) of the calcined powders were calculated by the X-ray line broadening technique performed on the (111) diffraction of  $\text{CeO}_2$  lattice using the Scherrer Equation 1:

$$D_{\text{XRD}} = K\lambda / \beta \cos \theta \quad (1)$$

where  $\theta$  is the Bragg angle of diffraction lines,  $K$  is a shape factor ( $K = 0.9$  in this work),  $\lambda$  is the wavelength of incident X-rays ( $\lambda = 0.15406$  nm) and  $\beta$  is the corrected half width given by Equation 2:

$$\beta^2 = \beta_m^2 - \beta_s^2 \quad (2)$$

where  $\beta_m$  is the measured half-width and  $\beta_s$  is the half width of a standard  $\text{CeO}_2$  sample with a known crystallite size of larger than 1  $\mu\text{m}$ .

Lattice parameter of synthesized powders was calculated by Equation 3:

$$d_{(hkl)} = \frac{a}{\sqrt{h^2 + k^2 + l^2}} \quad (3)$$

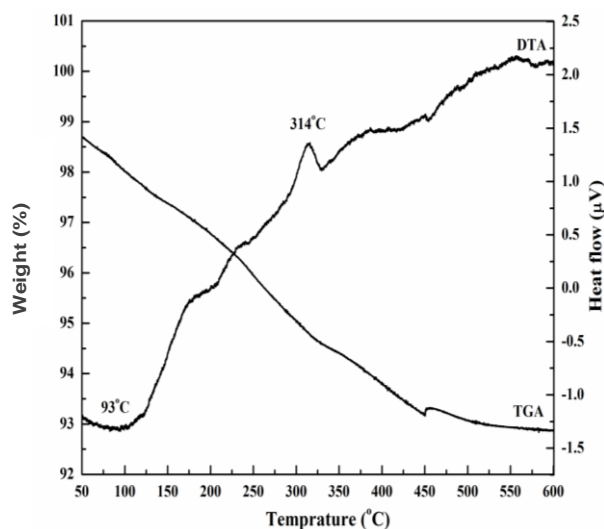
Differential thermal analysis/thermogravimetry (DTA/TGA) of the dried precursors was done on a simultaneous thermal analysis (STA) analyzer (PL STA1640) in flowing air with a heating rate of  $10^\circ\text{C min}^{-1}$ . The  $\alpha$ -Alumina was used as the reference material. Particle morphologies of the  $\text{CeO}_2$  powders were observed via scanning electron microscopy (SEM) (Stereo Scan S360 and WEGA TE Scan). The microstructure, size distribution and shape of the particles were investigated with transmission electron spectroscopy (TEM) (Jeol USA and Zeiss Germany). Further structural characteristics were determined by FT-IR spectroscopy (Bruker vector33).

## RESULTS AND DISCUSSION

Figure 1a and b show the powder XRD pattern of the synthesized powders after calcination at 600 and 800°C for 2 h, respectively. The diffraction peaks of all the samples assigned to cubic fluorite  $\text{CeO}_2$  are consistent with the JCPDS file of  $\text{CeO}_2$  (JCPDS No. 34-0394). The patterns in Figure 1a shows that increase in the aging time causes increase in the crystallinity. According to Figure 1b, at higher calcination temperature (800°C), the peaks are appreciably sharpened which indicate higher crystallinity. Also, Table 1 shows that increase in both the aging time and calcination temperature leads to appreciable grain growth in the crystallite size.

**Table 1.** Crystallite sizes of CeO<sub>2</sub> powders before and after calcinations.

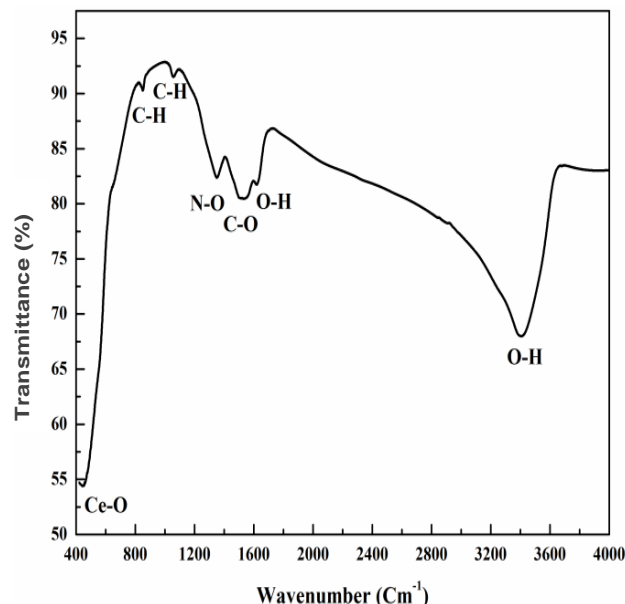
Calcination temperature (°C)	Powder name	Crystallite size (nm) from (111) peak	Lattice parameter (nm) from (111) peak
600	H <sub>1</sub>	5.9	0.5422
600	H <sub>2</sub>	6.5	0.5420
600	H <sub>3</sub>	7.3	0.5419
800	H <sub>2</sub>	24.4	0.5416
800	H <sub>3</sub>	26.8	0.5414

**Figure 2.** STA curves of H<sub>2</sub>.

Furthermore, it can be seen that in smaller crystallite sizes, due to the presence of more point defect, the lattice parameter became greater and the expansion occurred in the crystal lattice.

Figure 2 shows the DTA-TGA curves of the co-precipitated CeO<sub>2</sub> precursor which was aged for 1 h. It is seen that the TGA and STA curves illustrate two stages of the calcination process; the endothermic reaction at around 93°C and the exothermic reaction at ~314°C, due to the removal of the absorbed water and decomposition of residual nitrate, respectively (Godinho et al., 2007). In fact, STA study did not show significant processes occurring during the heating of the samples, which indicates that the formation of CeO<sub>2</sub> crystallites has already been taking place during precipitation and aging process.

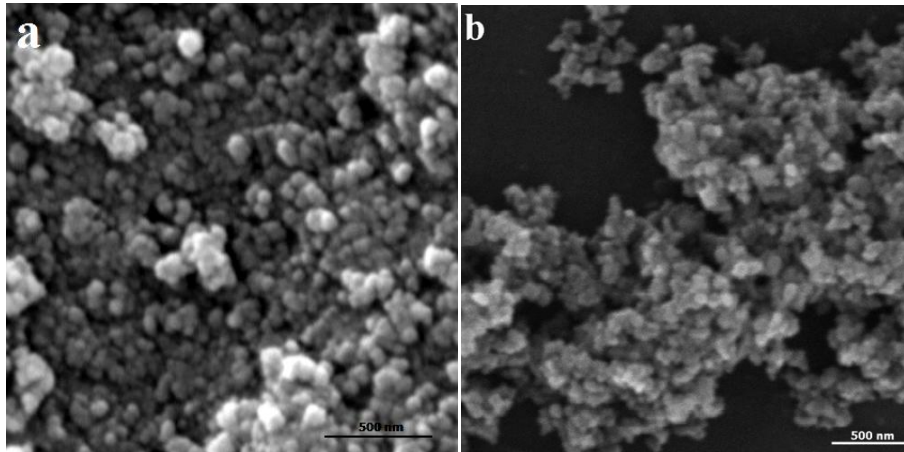
Figure 3 displays the FT-IR spectrum of CeO<sub>2</sub> nanoparticles in the wave number range from 400 to 4000 cm<sup>-1</sup>. The absorption bands around 3400 and 1620 cm<sup>-1</sup> are attributed to the stretching mode of water and hydroxyls, respectively. A sharp band at 1385 cm<sup>-1</sup> is indicative of N-O stretching vibration due to the traces of nitrate. The band at wave number below 500 cm<sup>-1</sup>

**Figure 3.** FT-IR spectra of CeO<sub>2</sub> nanoparticles with 1 h aging.

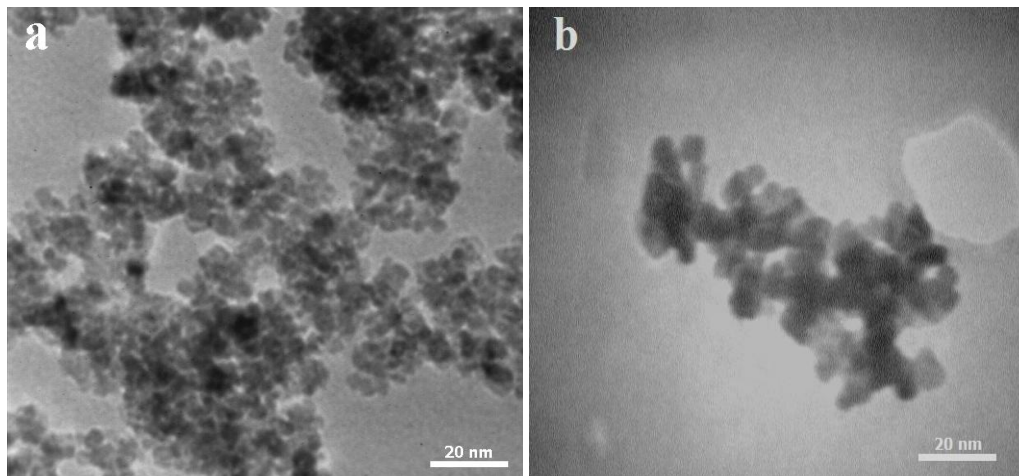
corresponds to the Ce-O bands (Athawale et al., 2009; Yue and Phanichphant, 2009).

Typical morphologies and agglomeration states of the nanopowders were obtained after 1 and 6 h aging observed as shown in Figure 4. It can be seen that the size of the secondary particles are ~50 and ~40 nm for H<sub>1</sub> (Figure 4a) and H<sub>2</sub> (Figure 4b) samples, respectively, consisting of the fine particles smaller than 10 nm. The SEM observations indicate increase in the results in the smaller nanoparticles with weaker agglomeration.

The crystalline size has also been directly observed using a transmission electron microscope (TEM). As shown in Figure 5, the calcined nanoparticles possess approximately spherical shape in the both of H<sub>1-600</sub> and H<sub>2-600</sub> samples. Moreover, the size distribution of the particles is uniform and the average crystallite size is ~3 and ~6 nm for H<sub>1</sub> and H<sub>2</sub>, respectively, which approximately agrees with the result obtained by XRD studies, indicating a well-defined dispersing state without significant hard agglomeration.



**Figure 4.** SEM photographs of  $\text{CeO}_2$  powders calcined at  $600^\circ\text{C}$ , (a)  $\text{H}_1$  and (b)  $\text{H}_2$ .



**Figure 5.** TEM photographs of  $\text{CeO}_2$  powders calcined at  $600^\circ\text{C}$ , (a)  $\text{H}_1$  and (b)  $\text{H}_2$ .

## Conclusion

In this work, ceria nanoparticles were synthesized by simple and cost-effective reverse co-precipitation method. Effect of aging time and calcination temperature was studied, which showed that aging time influences the size and morphology of the synthesized nanoparticles, that is, increase in aging time led to larger crystallite sizes, smaller particle sizes, weaker agglomeration and more crystallinity. Also, increase in the calcination temperature resulted in higher crystallinity and much coarser crystallites.

## REFERENCES

- Athawale AA, Bapat MS, Desai PA (2009). Hydroxide directed routes to synthesize nanosized cubic ceria ( $\text{CeO}_2$ ). *J. Alloys. Compd.*, 484: 211–217.
- Chandradass J, Nam B, Kim KH (2009). Fine tuning of gadolinium doped ceria electrolyte nanoparticles via reverse microemulsion process. *Colloids Surf A: Physicochem. Eng. Asp.*, 348: 130–136.
- Chevire F, Munoz F, Baker CF, Tessier F, Larcher O, Boujday S, Colbeau-Justin C, Marchand R (2006). UV absorption properties of ceria-modified compositions within the fluorite-type solid solution  $\text{CeO}_2\text{-Y}_6\text{WO}_{12}$ . *J. Solid State Chem.*, 179: 3184–3190.
- Corradi AB, Bondioli F, Ferrari AM, Manfredini T (2006). Synthesis and characterization of nanosized ceria powders by microwave-hydrothermal method. *Mater. Res. Bull.*, 41: 38–44.
- Dos Santos ML, Lima RC, Riccardi CS, Tranquilin RL, Bueno PR, Varela JA, Longo E (2008). Preparation and characterization of ceria nanospheres by microwave-hydrothermal method. *Mater. Lett.*, 62: 4509–4511.
- Godinho MJ, Gonçalves RF, Santos LPS, Varela JA, Longo E, Leite ER (2007). Room temperature co-precipitation of nanocrystalline  $\text{CeO}_2$  and  $\text{Ce}_{0.8}\text{Gd}_{0.2}\text{O}_{1.9-8}$  powder. *Mater. Lett.*, 61: 1904–1907.
- Gorte RJ, Zhao S (2005). Studies of the water-gas-shift reaction with ceria-supported precious metals. *Catal. Tod.*, 104: 18–24.
- Hassanzadeh-Tabrizi SA, Mazaheri M, Aminzare M, Sadrezaad SK (2010). Reverse precipitation synthesis and characterization of  $\text{CeO}_2$  nanopowder. *J. Alloys Compd.*, 491: 499–502.
- Izu N, Shin W, Murayama N (2003). Fast response of resistive-type oxygen gas sensors based on nano-sized ceria powder. *Sens.*

- Actuators B., 93: 449–453.
- Jasinski P, Suzuki T, Anderson HU (2003). Nanocrystalline undoped ceria oxygen sensor. *Sens. Actuators B.*, 95: 73–77.
- Lapa CM, de Souza DPF, Figueiredo FML, Marques FMB (2010). Two-step sintering ceria-based electrolytes. *Int. J. hydrogen energy*, 35: 2737–2741.
- Li J, Ikegami T, Wang Y, Mori T (2002). Nanocrystalline  $Ce_{1-x}Y_xO_{2-x/2}$  ( $0 \leq x \leq 0.35$ ) Oxides via Carbonate Precipitation: Synthesis and Characterization. *J. Solid State Chem.*, 168: 52–59.
- Llugar M, Vitaskova L, Sulcova P, Tena MA, Badenes JA, Monros G (2010). Red ceramic pigments of terbium-doped ceria prepared through classical and non-conventional coprecipitation routes. *J. Euro. Cer. Sci.*, 30: 37–52.
- Matovic B, Boskovic S, Logar M, Radovic M, Dohcevic-Mitrovic Z, Popovic ZV, Aldinger F (2010). Synthesis and characterization of the nanometric Pr-doped ceria. *J. Alloys Compd.*, 505: 235–238.
- Matovic B, Dohcevic-Mitrovic Z, Radovic M, Brankovic Z, Brankovic G, Boskovic S, Popovic ZV (2009). Synthesis and characterization of ceria based nanometric powders. *J. Power Sour.*, 193: 146–149.
- Mukherjee ST, Bedekar V, Patra A, Sastry PU, Tyagi AK (2008). Study of agglomeration behavior of combustion-synthesized nanocrystalline ceria using new fuels. *J. Alloys Compd.*, 466: 493–497.
- Souza ECC, Brito HF, Muccillo ENS (2010). Optical and electrical characterization of samaria-doped ceria. *J. Alloys Compd.*, 491: 460–464.
- Tao Y, Shao J, Wang J, Wang WG (2009). Morphology control of  $Ce_{0.9}Gd_{0.1}O_{1.95}$  nanopowder synthesized by sol-gel method using PVP as a surfactant. *J. Alloys Compd.*, 484: 729–733.
- Tok AIY, Boey FYC, Dong Z, Sun XL (2007). Hydrothermal synthesis of  $CeO_2$  nano-particles. *J. Mater. Proc. Technol.*, 190: 217–222.
- Truffault L, Ta M, Devers T, Konstantinov K, Harel V, Simmonard C, Andrezza C, Nevirkovets IP, Pineau A, Veron O, Blondeau JP (2010). Application of nanostructured Ca doped  $CeO_2$  for ultraviolet filtration. *Mater. Res. Bull.*, 45: 527–535.
- Uekawa N, Ueta M, Wu YJ, Kakegawa K (2002). Characterization of  $CeO_2$  Fine Particles by Homogeneous Precipitation Method with Polyethylene Glycol. *Chem. Lett.*, 8: 854–855.
- Xin J, Cai W, Tichy JA (2010). A fundamental model proposed for material removal in chemical-mechanical polishing. *Wear*, 268: 837–844.
- Yamashita M, Kameyama K, Yabe S, Yoshida S, Fujishiro Y, Kawai T, Sato T (2002). Synthesis and microstructure of calcia doped ceria as UV filters. *J. Mater. Sci.*, 37: 683–687.
- Yao SY, Xie ZH (2007). Deagglomeration treatment in the synthesis of doped-ceria nanoparticles via coprecipitation route. *J. Mater. Proc. Technol.*, 186: 54–59.
- Zec S, Boskovic S, Kaluderovic B, Bogdanov Z, Popovic N (2009). Chemical reduction of nanocrystalline  $CeO_2$ . *Cer. Int.*, 35: 195–198.
- Zhou XD, Huebner D, Anderson HU (2002). Size-induced lattice relaxation in  $CeO_2$  nanoparticles. *Appl. Phys. Lett.*, 80: 3814–3816.